

Eigenstate-analysis using Sakurai-Sugiura method with O(N)-DFT code CONQUEST

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Today's topics

- Electronic structure analysis of large materials using Sakurai-Sugiura (SS) method and CONQUETS

Introduction of CONQUEST

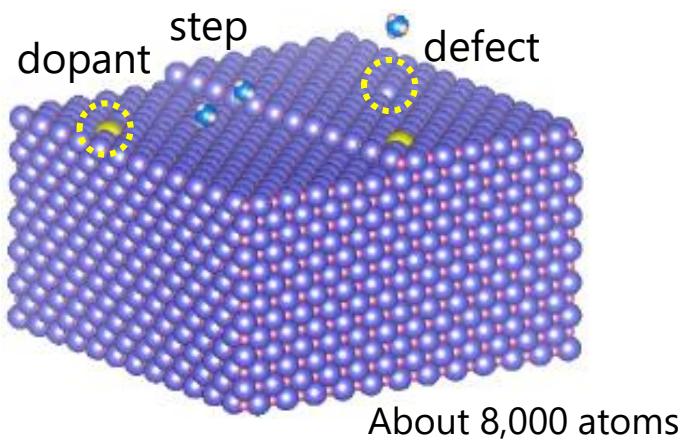
O(N) DFT

Multi-site method

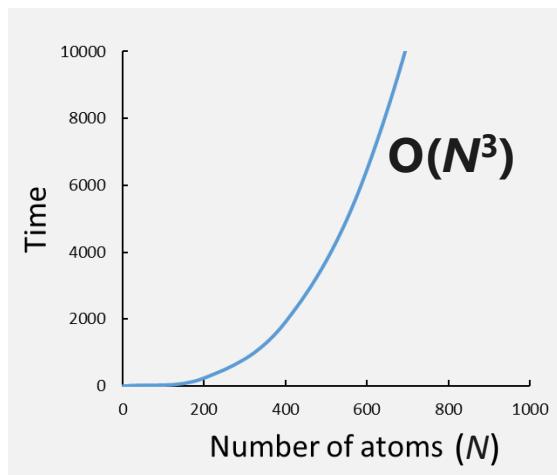
O(N)-DFT + SS

Multi-site method + SS

Large-scale DFT calculations



Large-scale model is needed
to represent **specific**
atomic and electronic structures



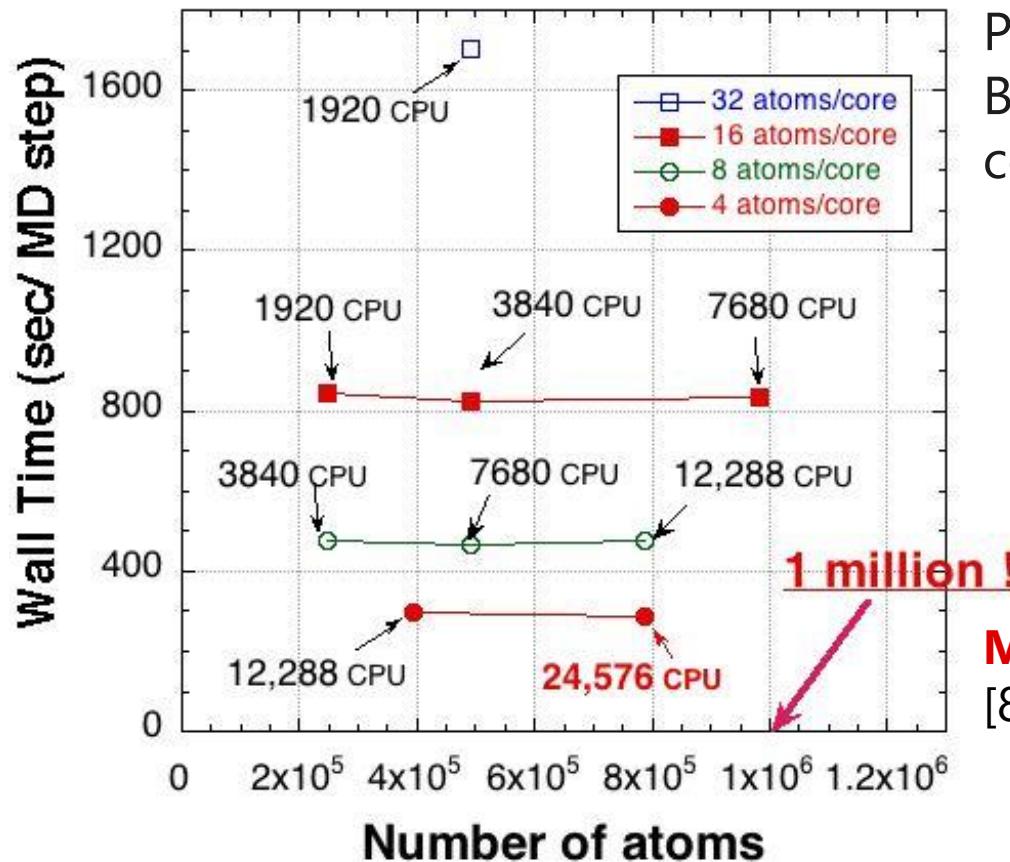
Scales cubically to
system size (# of atoms N) [**$O(N^3)$**]

Usually **$\leq 1,000$ atoms**

Large-scale DFT with CONQUEST

Concurrent **O(N)** QUantum Electronic Structure Technique [1]

DFT code for large-scale systems



Parallel efficiency in K-computer
Bulk Si,
constant # of atoms / node

Million atom systems: 5 min./MD step
[88,128 CPU = 705,024core]

Large-scale DFT with CONQUEST

Concurrent **O(N)** **QU**antum **E**lectronic **S**tructure **T**echnique [1]

DFT code for large-scale systems

- Local orbital functions
- Order- N method (Density matrix minimization)
- High parallel efficiency

Local orbital functions

Construct from basis functions

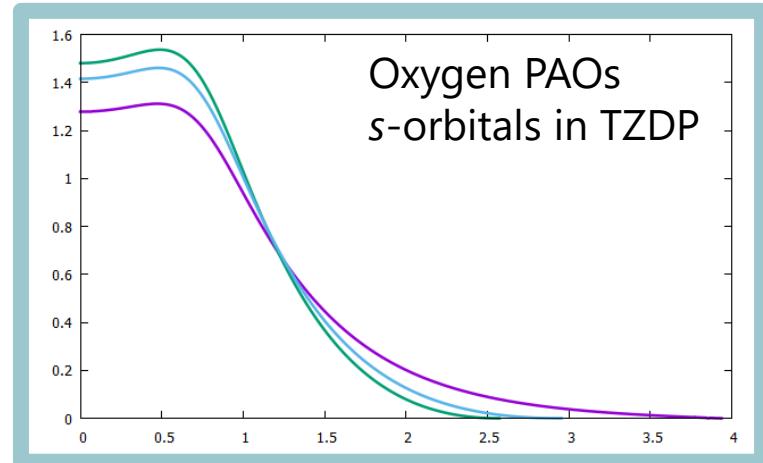
$$\phi_{\alpha}(r) = \sum_{\mu} c_{\alpha\mu} \chi_{\mu}(r)$$

Local orbitals

(Support function=SF)

Real-space

⋮



- Finite-element (B-spline) basis (akin to plane-wave functions)
- **Pseudo atomic orbital (PAO) basis**

$$\chi_{\mu}(r) = \chi_i^{\zeta lm}(r) = R_i^{l\zeta}(r) Y_i^{lm}(\mathbf{r})$$

ζ ... Radial functions
 l ... Angular momentum number
 m ... Magnetic quantum number

Radial functions Spherical harmonic functions

How to optimize electronic density

- Diagonalization

$$\mathbf{H}\mathbf{C} = \boldsymbol{\varepsilon}\mathbf{S}\mathbf{C}$$

$\mathcal{O}(N^3)$

\mathbf{H} : Electronic Hamiltonian

\mathbf{S} : Overlap matrix

\mathbf{C} : Eigenvectors (Bands/MOs)

$\boldsymbol{\varepsilon}$: Eigenvalues (Band/MO energies)

by using Scalapack

- Density matrix minimization → **$\mathcal{O}(N)$**

O(N) in CONQUEST

- **Density matrix minimization (DMM)**

$$E_{\text{tot}} = E_{\text{T}} + E_{\text{ps}} + E_{\text{H}} + E_{\text{xc}} + E_{\text{M}}$$

$$E_{\text{T}} = -\frac{\hbar^2}{m} \int d\mathbf{r} \left[\nabla^2 \rho(\mathbf{r}, \mathbf{r}') \right]_{\mathbf{r}=\mathbf{r}'}$$

$$\rho(\mathbf{r}, \mathbf{r}') = \sum_i f_i \psi_i(\mathbf{r}) \psi_i^*(\mathbf{r}')$$

$$E_{\text{ps}} = 2 \int d\mathbf{r} d\mathbf{r}' V_{ps}(\mathbf{r}', \mathbf{r}) \rho(\mathbf{r}, \mathbf{r}')$$

$$= \sum_{\alpha\beta} \phi_\alpha(\mathbf{r}) K_{\alpha\beta} \phi_\beta^*(\mathbf{r}')$$

$$E_{\text{H}} = \frac{1}{2} \int d\mathbf{r} d\mathbf{r}' n(\mathbf{r}) n(\mathbf{r}') / |\mathbf{r} - \mathbf{r}'| \quad n(\mathbf{r}) = 2\rho(\mathbf{r}, \mathbf{r})$$

$$E_{\text{xc}} = \int d\mathbf{r} n(\mathbf{r}) e_{\text{xc}}[n(\mathbf{r})]$$

$$\frac{\partial E^{\text{tot}}}{\partial L} = 6(SLH + HLS)_{\alpha\beta} - 4(SLSLH + SLHLS + HLSLS)_{\alpha\beta}$$

L = Auxiliary density matrix

$O(N)$ in CONQUEST

- Density matrix minimization (DMM)

$$\frac{\partial E}{\partial L}$$



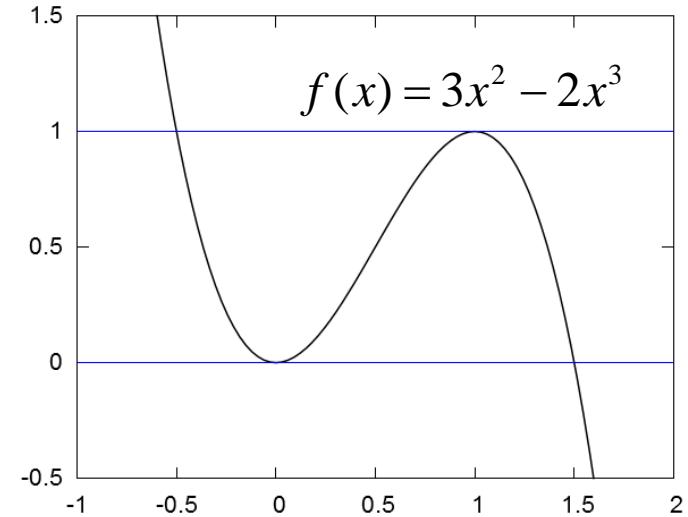
**Energy minimization
keeping idempotency condition (LNV)**

$$K = 3LSL - 2LSLSL$$

K = Density matrix

L = Auxiliary density matrix

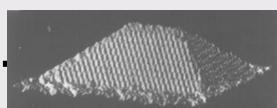
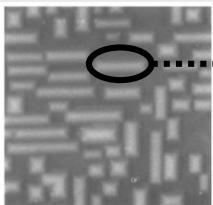
S = Overlap matrix



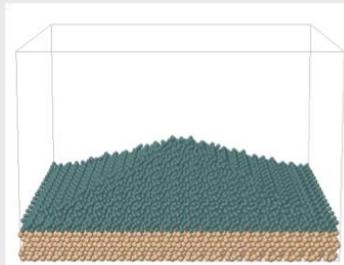
Spacial cutoff for $L \Rightarrow O(N)$

Capability of CONQUEST

Geometry optimization



STM of Ge hut cluster on Si surface
[Y.-W. Mo et al. PRL 65, 1020 (1990).]



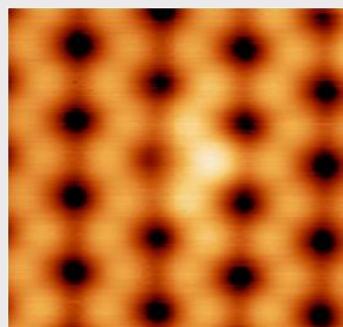
About **200,000 atoms**

Optimized geometry by CONQUEST
[Miyazaki et al., JPSJ, 77, 123706 (2008).]

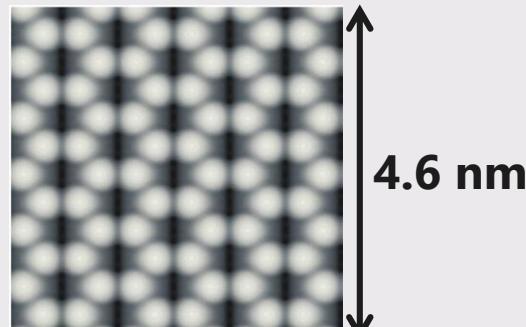
STM, STS

STM picture of Si (100) surface (-1.5eV)

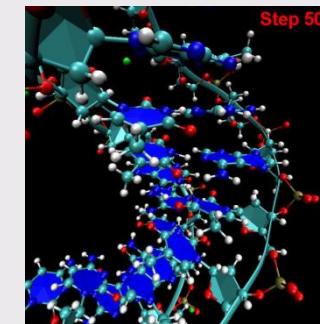
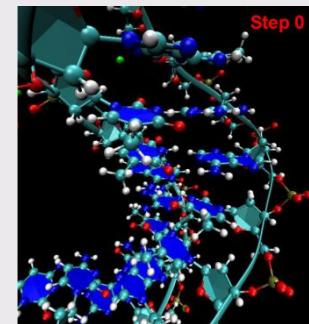
Exptl



CONQUEST



Ab initio MD



Hydrated DNA (**3,439 atoms**) PBE/SZP
[T. Otsuka et al, in preparation.]

Si (**32,768 atoms**) LDA/DZP
Combination with extended Lagrangian method
[M. Arita et al, JCTC, 10, 5419 (2014).]

**TDDFT,
Constraint DFT, EXX,
Blue-moon ensemble
simulation,**

...

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Multi-site method

O(N)-DFT + SS

Multi-site method + SS

Eigenstates with $O(N)$ in CONQUEST

$O(N)$ method

- 😊 Low cost: $O(N)$
- 😊 Energies and geometries
- 😐 Metals??
- 😢 **No 1-e⁻ wave functions (Bands, MOs)**



Electronic states:

Orbital pictures
Density of states
STM images ...
(IPs, EAs, excitation energies)

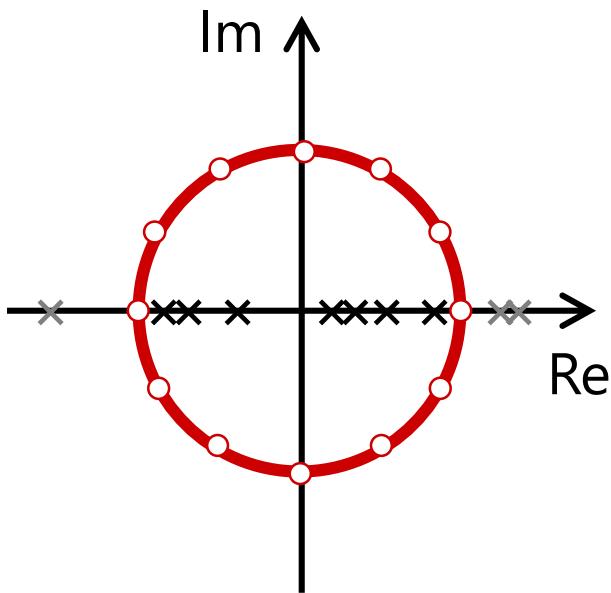
Sakurai-Sugiura method:

Projection method to obtain

eigenstates in a finite range

Sakurai-Sugiura method

Projection method to obtain the eigenstates in a **finite range**



(1) **N** simultaneous linear equations
(number of integration points on the curve)

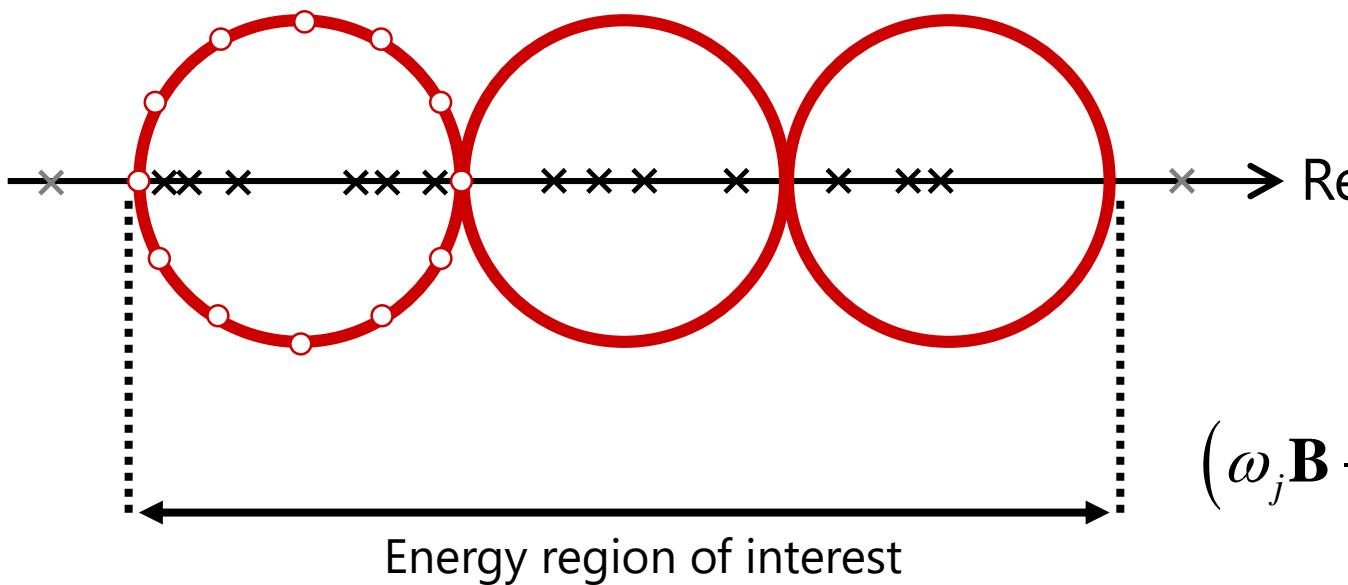
$$(\omega_j \mathbf{B} - \mathbf{A}) \mathbf{y}_j = \mathbf{v} \quad \omega_k = \gamma + R_D \exp\left(\frac{2\pi i}{N} k\right), \quad (k = 0, 1, \dots, N-1)$$

(2) **m × m** diagonalization
(number of states in the region) $\mathbf{H}_m^< - \lambda \mathbf{H}_m$

$$\mathbf{H}_m := \begin{pmatrix} \mu_0 & \mu_1 & \cdots & \mu_{m-1} \\ \mu_1 & \mu_2 & \cdots & \mu_m \\ \vdots & \vdots & \ddots & \vdots \\ \mu_{m-1} & \mu_m & \cdots & \mu_{2m-2} \end{pmatrix} \quad \mathbf{H}_m^< := \begin{pmatrix} \mu_1 & \mu_2 & \cdots & \mu_m \\ \mu_2 & \mu_3 & \cdots & \mu_{m+1} \\ \vdots & \vdots & \ddots & \vdots \\ \mu_m & \mu_{m+1} & \cdots & \mu_{2m-1} \end{pmatrix}$$

$$\mu_k = \frac{1}{2\pi i} \int_{\Gamma} \frac{f(z)}{(z - \gamma)^{k+1}} dz \quad \mu_k \approx \hat{\mu}_k := \frac{1}{N} \sum_{j=0}^{N-1} (\omega_j - \gamma)^{k+1} f(\omega_j), \quad (k = 0, 1, \dots) \quad f_j \leftarrow \mathbf{u}^H \mathbf{y}_j, \quad (j = 0, \dots, n-1)$$

Sakurai-Sugiura method



N_{SS} = number of quadrature points

M_{SS} = number of complex moments

L_{SS} = number of columns of the source matrix \mathbf{V} ($M_{\text{SS}}L_{\text{SS}} > m$)

Estimate the number of eigenstates in the region stochastically by using Sylvester's law of inertia

Procedure

Construct Hamiltonian and overlap matrices with **CONQUEST**



(sum over k-points)

(save only lower-triangle elements, larger than threshold)

Eigenstate calculation with Sakurai-Sugiura calculations
with **zPares**

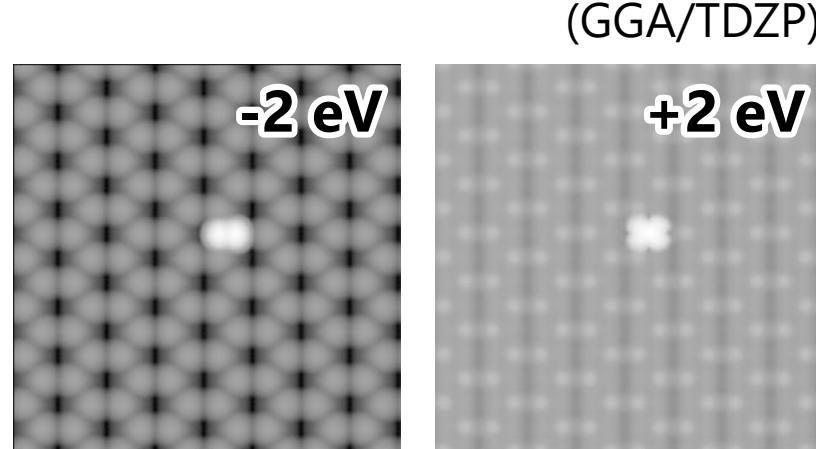
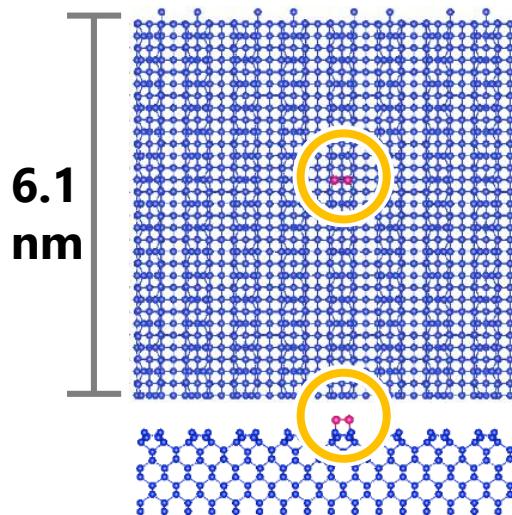


Read eigenvectors and eigenvalues with **CONQUEST**

MO / Band information → Electronic properties
STM pictures

STM images

P₂ on Si(100)



Scalapack (PDSYGVX)



Sakurai-Sugiura
($N_{SS}=16$, $M_{SS}=8$, $L_{SS}=100$)

4 node
5,064 s



924 s

64 node
629 s



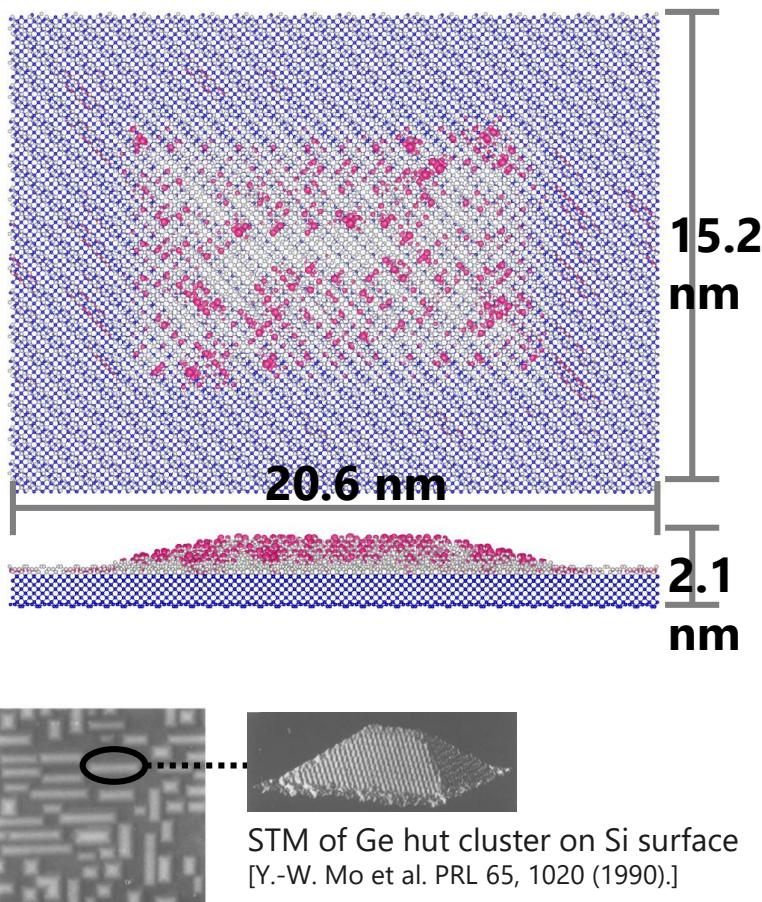
85 s

by PACS-IX (Intel Xeon E5-2670v2) in Univ of Tsukuba

Large systems

Ge hut cluster on Si(001)

- Charge distributions



$\pm 0.01\text{eV}$ around Fermi level

213,633 states



43 states (GGA/SZP)

atoms	node	time
23,737	64	146
194,573	6,400	2,399

($N_{\text{SS}}=16$, $M_{\text{SS}}=8$, $L_{\text{SS}}=64$)

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Large-scale DFT with CONQUEST

Concurrent **O(N)** **QU**antum **E**lectronic **S**tructure **T**echnique [1]

DFT program for large scale systems

- High parallel efficiency
- Order- N method (Density matrix minimization)
- **Local orbital function (support function)**

Localized in finite region around each atom → sparse

Cost scales cubically to the number of functions

$$\begin{array}{ll} \text{Diagonalization} & aN^3 \\ \text{Order-}N & bN \end{array}$$

⋮

N = number of atoms

Multi-site functions

No contraction

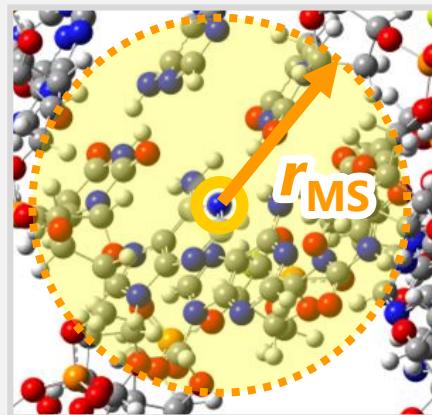
Primitive functions

Single-site contraction

$$\phi_{i\alpha}(r) = \sum_{\varsigma \in i} c_{i\alpha}^{\varsigma lm} \chi_i^{\varsigma lm}(r)$$

Linear combination of AOs on each atom

Multi-site contraction



$$\phi_{i\alpha}(r) = \sum_k^{\text{neighbors}} \sum_{\varsigma lm \in k} C_{i\alpha}^{k\varsigma lm} \chi_k^{\varsigma lm}(r)$$

Linear combination of AOs in cutoff region

Contract to be **minimal size**
keeping primitive AOs' accuracy

Gradient wrt coefficients

$$\frac{\partial E}{\partial C_{i\alpha,k\mu}} = \frac{\partial E}{\partial \phi_{i\alpha}} \frac{\partial \phi_{i\alpha}}{\partial C_{i\alpha,k\mu}} = \frac{\partial E}{\partial \phi_{i\alpha}} \chi_{k\mu}$$

$$\boxed{\frac{\partial E}{\partial \phi_{i\alpha}(\mathbf{r})} = 4 \sum_{\beta} \left[K_{\alpha\beta} \hat{H} + G_{\alpha\beta} \right] \phi_{j\beta}(\mathbf{r})}$$

Diag. $G_{\alpha\beta} = \sum_n f_n \varepsilon_n u_{n\alpha} u_{n\beta}^*$

$$K_{\alpha\beta} = \sum_n f_n u_{n\alpha} u_{n\beta}^*$$

O(N) $G_{\alpha\beta} = 3(LHL)_{\alpha\beta} - 2(LSLHL + LHLSL)_{\alpha\beta}$

$$K_{\alpha\beta} = 3(LSL)_{\alpha\beta} - 2(LSLSL)_{\alpha\beta}$$

Advantages of multi-site method

- Applicable to metallic systems

Cutoff is introduced for support functions
but not for wave functions

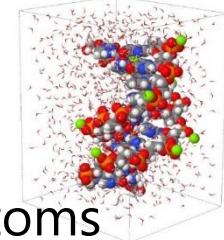
- Improve accuracy as much as we want

(ex) Si atom

	Primitive	Multi-site support functions
DZP	13	4
TZDP	22	4
		Always to be minimal size

Computational time

DNA in water



	Primitive AO (DZP)	Multi-site SFs (8 bohr)	Multi-site SFs (16 bohr)
# of functions	27883	4744	4744
Time [sec.]			
Matrix construction	28.0	34.5	484.8
Diagonalization	12317.4	627.7	566.3
Gradient of coefficients	—	4.3	25.7

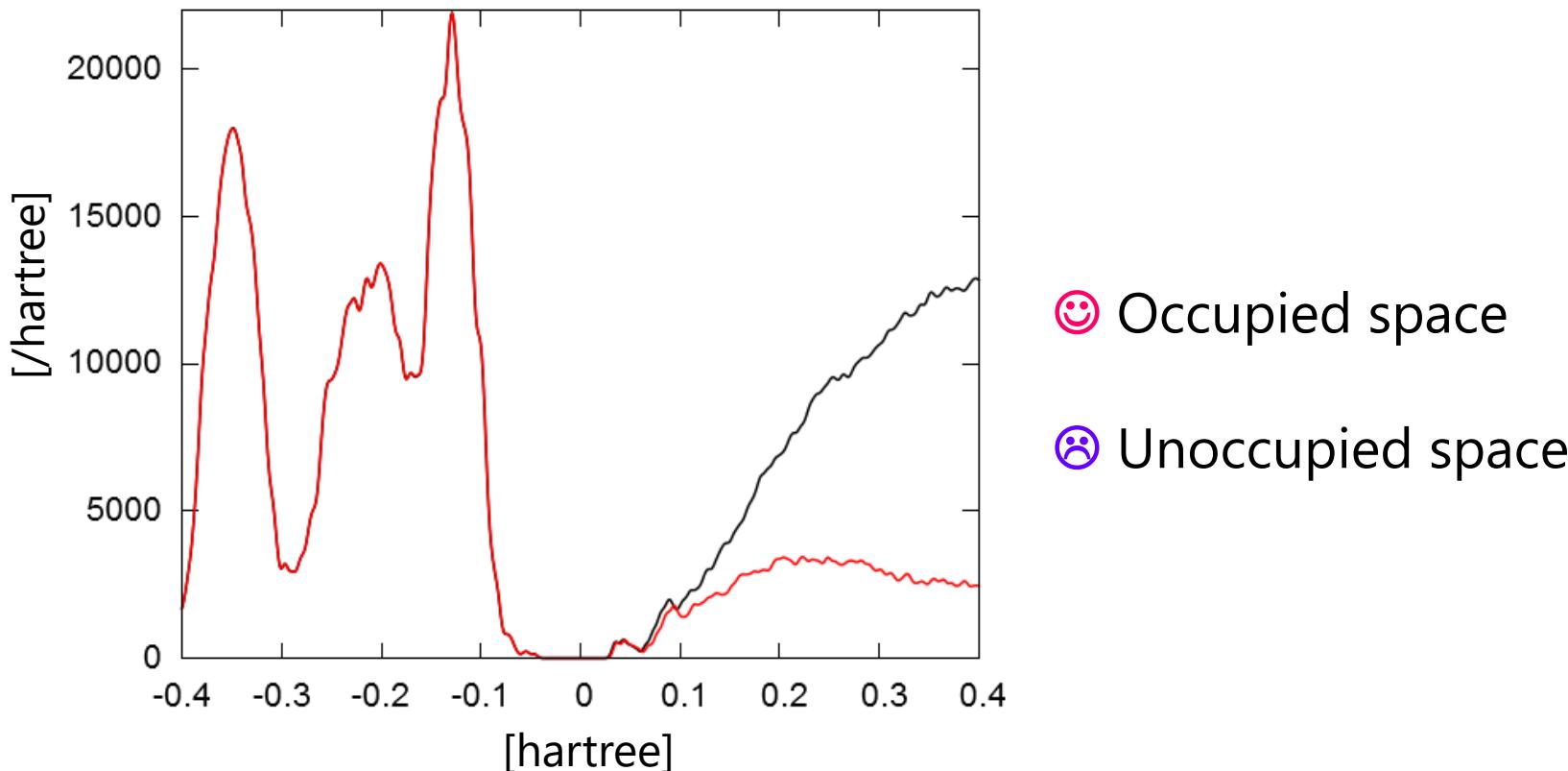
PBE / DZP
HA8000t (Kyushu Univ)
96 cores parallel

Large reduction! (about 1/12)

DOS of DNA in water (8 bohr)

Cutoff $r_{\text{MS}} = 8.0$ bohr

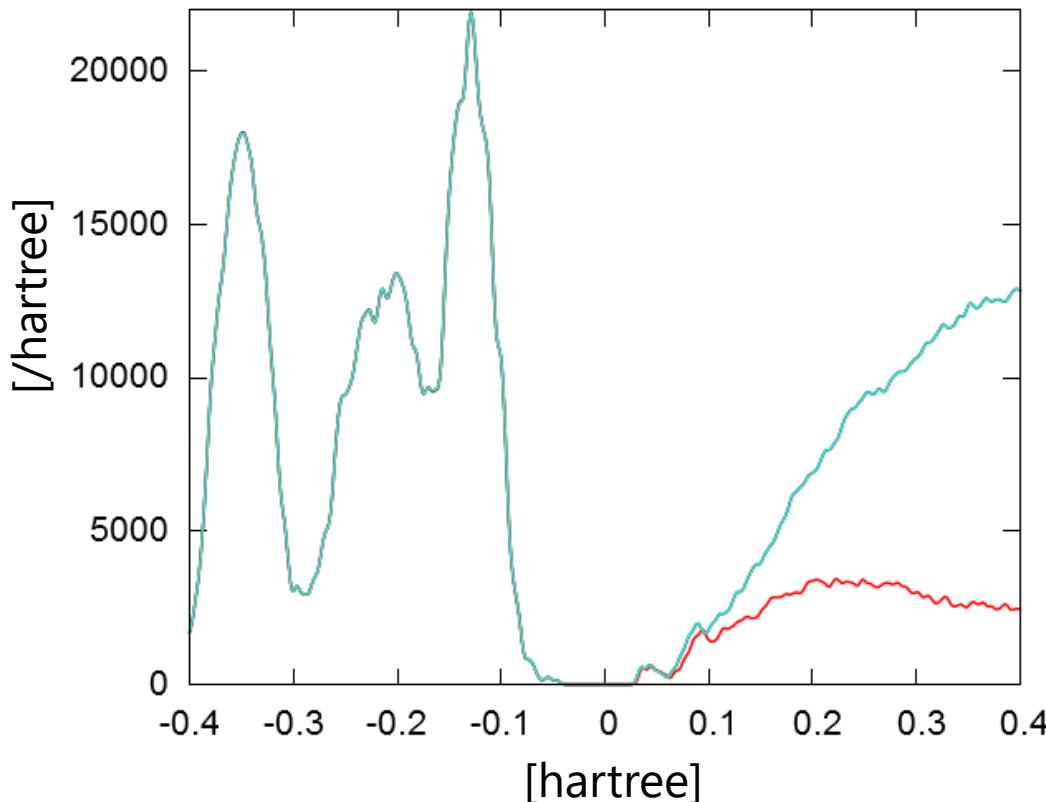
Black: Primitive AO (DZP)
Red : Multi-site SFs



DOS of DNA in water (8 bohr)

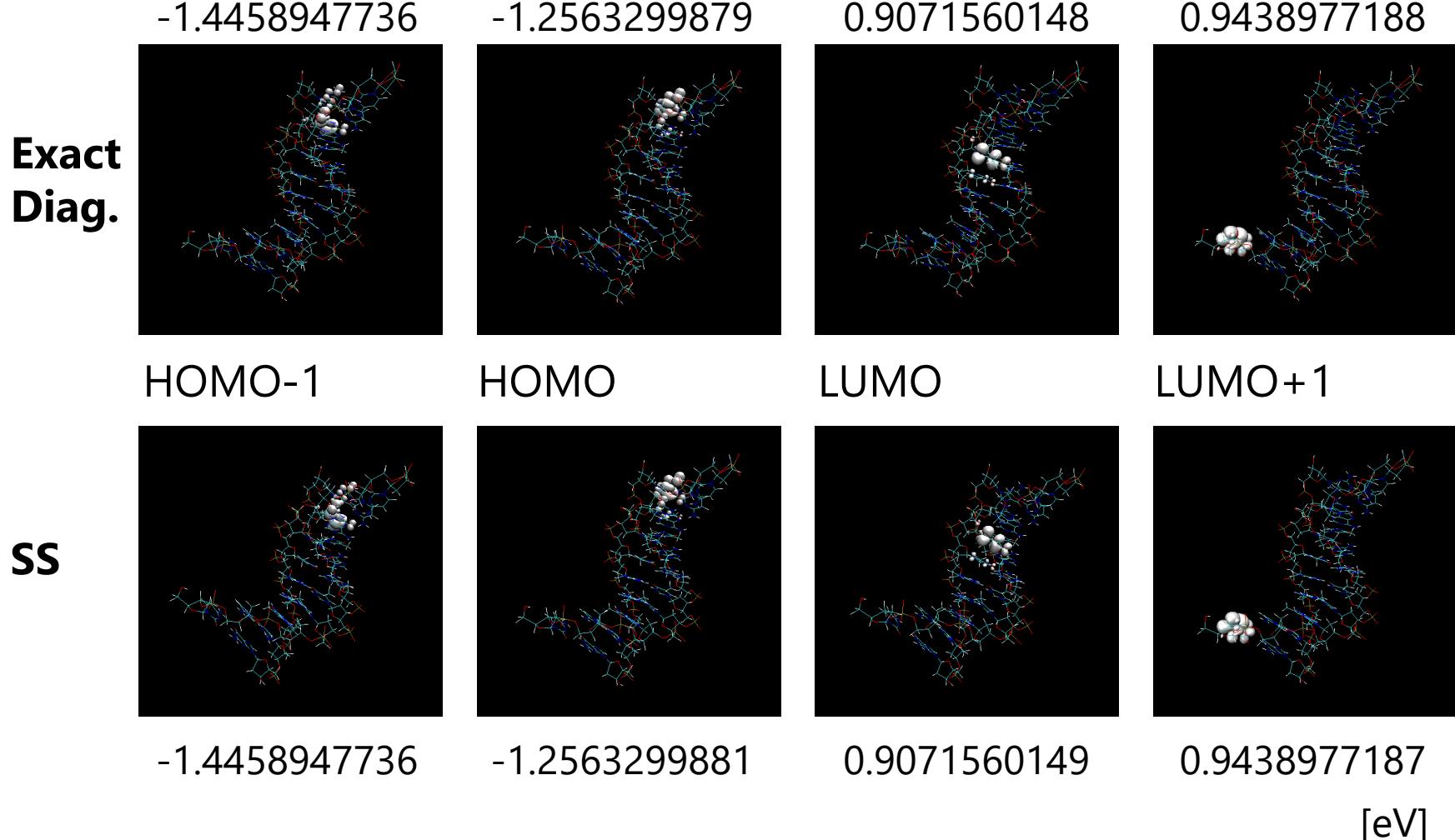
Cutoff $r_{\text{MS}} = 8.0$ bohr

Black: Primitive AO (DZP)
Red : Multi-site SFs
Blue : Primitive AO (DZP) with charge of multi-site SFs



One-shot recalculation
using multi-site charge
with primitive AOs
(by Sakurai-Sugiura method)

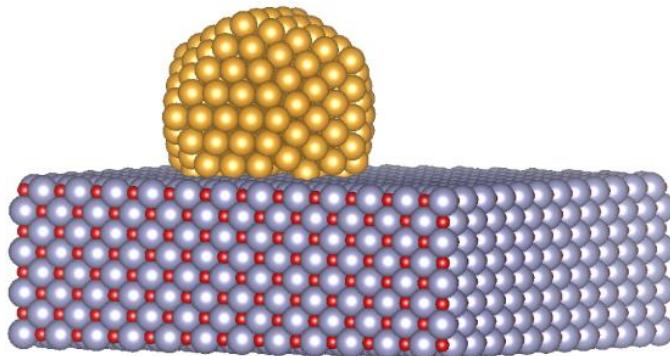
MOs of DNA in water



Summary

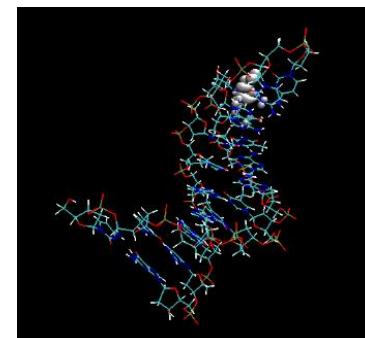
Investigation of eigenstates (1e- wave functions) becomes available even with $O(N)$ method by using Sakurai-Sugiura method

Multi-site method makes available accurate calculation with low computational cost simulation of large metallic systems



Metallic nanoparticles

HOMO



MO pictures of DNA

LUMO

